

Fig. 4. The hydrogen bonding of the barbital anion [see also Table 2(b)].

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# The Crystal Structure of Gallium Thiophosphate, GaPS<sub>4</sub>

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GaPS<sub>4</sub> is monoclinic with a=8.603 (4), b=7.778 (3), c=11.858 (5) Å,  $\beta=135.46^{\circ}$ , space group  $P_{2_1/c}$  and Z=4. The structure has been determined from 706 reflexions and refined to R=0.066. It consists of puckered hexagonally close-packed sulphur layers. Of the interlayers only every other one is occupied by Ga and P cations. Filled and empty interlayers thus alternate, causing perfect cleavability parallel to (100). Ga and P are each surrounded by four sulphur atoms at the corners of distorted tetrahedra.

#### Introduction

Gallium thiophosphate belongs to a group of ternary metal-phosphorus-sulphur-compounds of the type  $MPS_4$  (M=B, Al, Ga, In, Sb, Bi). The crystal structures of  $AIPS_4$  and  $BPS_4$  have been solved by Weiss & Schaefer (1960, 1963). The structure of  $InPS_4$  has been determined by Carpentier, Diehl & Nitsche (1970).

Studying the crystal growth of metal-phosphorussulphur compounds by vapour transport, Nitsche & Wild (1970) obtained single crystals of GaPS<sub>4</sub>. The crystals were grown by chemical transport with iodine in a temperature gradient from 650 to 600 °C. Cell constants and space group have been determined by Buck & Nitsche (1971). The platelike morphology of the crystals differs significantly from that of the other compounds mentioned above. It therefore seemed reasonable for a better understanding of the crystal chemistry of ternary phosphorus chalcogenides to determine also the crystal structure of GaPS<sub>4</sub>.

#### Crystal data

Least-squares refinement of powder data using the X-RAY 70 system (*PARAM*, Stewart, Kundell & Baldwin, 1970) gave the lattice constants listed in Table 1.

Table 1. Crystal data for gallium thiophosphate  $GaPS_4$ 

a = 8.603 (4)  Å b = 7.778 (3)	Space group:	$P2_1/c$ (No. 14) Z=4
c = 11.858 (5) $\beta = 135.46^{\circ} (4)$		$\rho_{exp} = 2.65 \text{ g cm}^{-3}$ $\rho_{calc} = 2.72 \text{ g cm}^{-3}$
$U = 556.54 \text{ Å}^3$		

Systematic extinctions: h0!: l = 2n + 10k0: k = 2n + 1

#### Intensity data

Integrated b-axis equi-inclination Weissenberg photographs (k=0, 1, 2, 3, 4) were collected by the threefilm technique with Cu K $\alpha$  radiation. The intensities were measured with a Jenoptik photometer (estimated average accuracy 7–8%). Low intensities were estimated visually by means of a calibrated scale. 631 independent reflexions were recorded. Additionally, 75 reflexions with intensities not differing significantly from the background were set to zero and were included as 'less-than's' in the calculations. Intensity data were corrected for Lorentz and polarization factors. The crystal specimen used was of a polyhedral shape that could be approximated by a sphere with a diameter of 0.23 mm. Therefore an absorption correction for that sphere ( $\mu R = 2.4$ ) was applied to the intensity data.

### Structure determination and refinement

The volume of the unit cell corresponds fairly well to the volume corresponding to a hexagonal close packing of 16 sulphur ions. The  $d_{(100)}$  value of 6.05 Å is twice the distance of two sulphur layers. Therefore it was assumed that the structure contains close-packed sulphur layers. The perfect cleavability parallel to (100) suggested that the cations Ga and P were not distribbuted evenly among the interstices between all sulphur layers. It seemed likely, rather, that empty and filled interlayers were alternating in the structure.

Patterson projections were computed with 0kl (42 reflexions) and h0l (82) intensities. The resolution of these Patterson maps was not very high but they confirmed the above model. The low resolution was not only due to the small number of reflexions used but also – as was seen later – to the fact that many Patterson vectors differed only slightly in direction and length.

With the atomic parameters derived from the Patter-

Table 2. Observed and calculated structure factors

6+0+1 e 507 -567 4 1949 2072 1. 144 101 3 493 491 10 510 500	10 10 95 12 34 371 14 203 241 0*0*6 2 153 -120 2 357 -352	-1+1+L 1 915 829 2 505 402 3 927 -935 • 556 570 5 580 567 5 580 567 7 -91 -719	+ 212 -102 5 3mm 302 5 255 7 3mm 235 7 3mm -389 5 174 230 7 560 538 17 -7 208 11 100 -201 12 001 261	-Artal 1 150 -100 2 04 41 3 123 -117 4 691 -44 5 266 -000 6 121 104 7 46 90	2:2:4 	- 732 670 -5+2+1 - 10 -178 - 30 -217 - 307 196 - 610 -596	12 115 120 13 c1 -70 0.3.L 1 91 -d5 2 103 82 3 256 221 4 30 - 220	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	-7,3,4 1 415 -414 2 135 -104 3 127 129 4 275 -240 5 149 -102 6 161 136 7 ( -14	7 373 361 6 510 542 7 0 26 10 0 20 2+4+L 7 242 216 140 210	-5,4, 1 235 2 157 3 172 4 493 5 331 6 ne
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2 402 337 4 723 540 0 125 -03 8 735 777 14 50 07 2+0+L 0 327 327	7+0+2 - +11 -+12 -7+0+2 - 2-03 -200 + 10+6 +100 - 77 - 40 -77	0 91 103 5 312 538 4 405 391 3 691 -710 2 527 253 1 885 903 1 353 254 -3.1*5	-3+1+L 1 330 -317 2 231 -170 3 651 715 4 524 -100 5 632 -100 5 173 204	-9+1+1 4 Tun tun 5 nut Tst 6 120 154 7 307 -710 9 00 -710 9 306 107 10 300 147	5 140 196 0 3 77 7 140 -129 1 594 -483 4 290 275 10 642 584 11 165 -141 12 362 -55	- 240 -257 - 3 -68 - 366 -326 -6.211 - 352 -794 - 236 -203	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	1 241 -214 2 761 -724 3 173 -145 4 0* 17 5 0* 39 6 274 -675 7 255 -254 6 33 -229 7 33 350	2 214 -171 3 45 46 - 200 238 404 -374 5 200 -254 7 257 227 3 64 52 9 304 -322 9 304 -322 10 133 134	1 374 559 2 103 41 3 -4238 4 1020 1000 5 515 355 6 00 -52 7 254 -267 6 362 -344 9 145 140	0 210 -++4+ 1 26# 2 5 42 5 70
0 nJ2 =074 4 J24 =200 2 471 =450 1 1043 =1502 -2+0+L 2 03 07 4 1917 =1075	3 1+43 -1510 10 177 81 12 1171 -1131 14 0+ 7  4 /14 770 577 -336	1 003 -907 2 025 -546 3 072 1036 4 527 -202 5 457 -802 n 71 -59 7 152 400 8 241 -200	7 010 044 4 715 -772 7 01 -1940 14 144 1844 11 467 445 12 444 446 13 419 -230 0+1+L	11 411 -430 12 168 -146 13 361 457 14 195 213 -1091+0 6 204 -221 7 396 437	3+2+L n 102 104 5 104 104 4 284 -281 5 05 02 2 07 -115 1 4 -75 3 284 -102	4 04 2 4 134 -133 4 007 -158 7 200 255 6 138 178 7 442 -177 11 150 162	-1+3+L 1 5-5 5+2 2 400 -430 3 545 -647 4 542 552 5 345 340 0 501 -512 7 510 -512	10 943 -551 11 200 -205 14 269 -215 15 269 -313 3+3+5 3 167 102 4 176 -737	11 242 241 12 271 -249 13 155 -117 14 416 413 -7+5+5 5 253 269 6 439 -395	10 192 -217 11 0+ -28 3+4+L 5 150 -156 4 714 698 3 453 429 2 314 -271	6 96 7 255 8 256 9 143 10 14 11 406 12 547 13 25P
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4 1123 1037 -3+0+C - 205 -004 - 205 -213 - 405 -400 - 105 -605 - 105 -605	0 519 517 10 306 907 10 307 907 10 307 907 10 307 907 10 300 937 10 230 230 10 250 227	4 0 45 3 313 302 2 193 -104 1 649 -314 0 021 -3.1*L 1 901 441	1 374 -303 + 251 351 - 374 617 - 223 -231 - 403 484 - 403 484 - 134 365 11 374 4365	1 234 -141 2 340 -111 3 845 -277 4 841 -44 5 246 -230 6 346 -230 6 346 -230 7 238 -424 7 238 -424 7 238 -424 7 238 -126	7 387 394 1 305 293 9 415 413 1 1270 -1390 11 179 171 12 14 445 44246	0 670 - 638 - 106 213 4 176 175 - 138 - 175 - 476 - 606 - 144 167 10 - 27 11 170 - 147 11 170 - 147 11 370 - 733	5 325 290 4 105 104 3 404 -643 2 003 -001 1 627 n33 0 321 340 -2+3+6	5 +++ ~550 0 54 551 7 561 330 0 54 449 9 500 -540 10 50 550 14 +50 370 12 0 30 14 -50 370 12 0 34	6 246 -247 4 64 -32 10 373 434 0+++L 1 325 -374 7 214 -147 3 775 787	5 3A2 -373 6 3+ -37 7 124 136 6 111 -153 9 317 -290 10 5+ -4A 11 73 65 4 4444	4 66 5 439 6 15 7 374 8 844 9 500 10 9 11 374
4404L 4004 - 2014 2014 - 2014 2014 - 2015 2014 - 2015 2014 - 2015	UFIFE 1 100 +119 2 32 +03 3 340 273 4 230 -193 3 212 -227 4 20 -99	2 12" -10" 3 550 -554 4 273 251 5 207 280 0 384 -347 7 7 4 -56 137 174 4 0 -22 10 193 -175	10 94 447 13 111 72 14 214 272 7+1+L -1 414 394 -7+1+L	10 407 -000 1+7+L A 134 -110 7 107 -175 6 00 -11 4 00 -20	5 160 160 4 233 205 5 103 -116 2 1240 1103 1 467 406 0 107 -106 -9124L	1" 105 108 -0.211 170 -138 1 50 70 2 244 205 144 -177 4 143 -197	1 442 -514 2 124 4247 3 572 nu3 4 54 -113 5 533 -514 5 933 544 7 237 203 5 152 143 9 430 -534	∟، 5 ہے۔ 1 1 م م ا 1 2 - م ا 1 2 - م ا 1 2 - م ا	4 1204 1191 5 476 -405 6 176 205 7 164 205 7 247 241 9 59 -50 1444L	4 239 -214 5 161 -147 2 269 235 1 434 305 3 04 32 -444L	13 304 -P+4: 167 4 513 5 461 6 140 7 257
- 541 555 - 475 496 - 501 455 - 501 455 - 501 455 - 501 555 - 505	/ 415 453 5 _n4 -240 7 1'0 -87 1+1+2 7 10/ 11n 5 _19 -191	11 0+ 75, 12 6+ -41 13 177 -104 4+1+L 5 236 241 4 6-39 3 12, -104	1 201 - 3° m 2 207 - 200 3 100 1100 4 67 - 74 5 104 - 24 5 74 7 124 134 7 124 134	3 240 -043 2 240 -117 1 152 -77 0 207 -147 -1474L 1 347 447 2 1427 1447	$\begin{array}{cccccccccccccccccccccccccccccccccccc$	- 76 108 - 112 139 - 124 -158 11 213 -147 11 22 -158 12 154 -158 12 154 -158 12 155 -118 14 378 -377	10 446 401 11 08 40 3+3+L 5 04 40 5 355 -353 4 94 -100 3 114 110	2 040 000 2 04 0 3 415 - 300 4 04 34 5 010 - 00 4 010 - 00 7 010 - 00 7 010 - 00 0 10 - 00 0 0 - 00 0 0 - 00 0 -		1 170 177 2 115 05 3 353 -369 4 55 57 5 275 265 7 362 -365 5 274 -285 9 274 -285	H 341 9 3.5 10 203 11 0 12 0 13 12 13 12 13 12
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son syntheses a structure factor calculation was performed using the atomic scattering factors for Ga, P and S (neutral atoms) given in *International Tables for X-ray Crystallography* (1968). No dispersion correction was applied. The isotropic temperature factor was kept at B=0. The resultant reliability index was R=0.27.

A full-matrix refinement of atomic coordinates and isotropic temperature factors was performed using pro-

# Table 3. Final atomic parameters and their standard deviations

gram CRYLSQ (X-RAY 70). No weighting scheme was applied to the observed reflexions. The reliability index converged to R=0.069 with zeros included and R=0.066 with them omitted. Observed and calculated structure factors are listed in Table 2. Final atomic coordinates and U values are given in Table 3.

A difference Fourier-map was featureless with the highest positive values smaller than  $\frac{1}{40}$  of the peaks in the Fourier-map.

#### **Results and discussion**

	x	у	Z	$U \times 10^2$
S(1)	0.2241(5)	0.0401 (6)	0.0103 (4)	1.38 (6)
S(2)	0.2516(5)	0.4795 (6)	0.0297 (4)	1.56 (7)
S(3)	0.6976 (5)	0.2683(6)	0.2048 (4)	1.44 (6)
S(4)	0.7484 (5)	0.7732 (6)	0.2078 (4)	1.63 (7)
Р	0.3586 (5)	0.2590 (6)	0.0103 (4)	1.04 (6)
Ga	0.6240 (2)	0.5369 (3)	0.2354(2)	1.82(3)

Fig. 1 shows the crystal structure of GaPS<sub>4</sub>. All atoms occupy position 4(e) at  $(x, y, z; \bar{x}, \bar{y}, \bar{z}; x, \frac{1}{2} + y, \frac{1}{2} - z; x, \frac{1}{2} - y, \frac{1}{2} + z)$  with x, y, z given in Table 3. The structure can be derived from a hexagonal close-packed arrangement of sulphur atoms. The sulphur layers are stacked normally to (100). Of the interlayers only every other



Fig. 1. The crystal structure of  $GaPS_4$ . (a) View normal to [100]; (b) view along the b axis.

one is occupied by cations [Fig. 1(b)]. Filled and empty interlayers thus alternate, causing the perfect cleavability. Ga and P are surrounded each by four sulphur atoms at the corners of distorted tetrahedra. The coordination tetrahedra of Ga and P are neighbouring and are linked together by edge-sharing. Because of this cation distribution the sulphur layers are slightly puckered and not planar as in ideal hexagonal close packing. The x coordinates of the sulphur atoms vary from 0.22 to 0.30 (0.25 in ideal case.)

In Fig. 1(a) (lower left) a gallium-sulphur and a phosphorus-sulphur tetrahedron are drawn. Table 4 gives the related interatomic distances and angles.

#### Table 4. Interatomic distances and angles in distorted gallium-sulphur and phosphorus-sulphur tetrahedra

Ga - S(1'')	2·275 (7) Å	S(1'')-Ga-S(2)	117.1(3)
Ga - S(2)	2.297 (5)	S(1'')-Ga-S(3)	105.0 (2)
Ga - S(3)	2.282(7)	S(1'') - Ga - S(4)	107.4(2)
Ga - S(4)	2.266(7)	S(2) - Ga - S(3)	89.3 (2)
S(1'') = S(2)	3.901 (8)	S(2) - Ga - S(4)	116.7(2)
S(1'') = S(3)	3.615(10)	S(3) - Ga - S(4)	120.5 (3)
S(1'') - S(4)	3.661(10)	S(1'') - S(4) - S(3)	56.6 (2)
S(2) - S(3)	3.218(8)	S(3) - S(1'') - S(4)	65.7(2)
S(2) - S(4)	3.884(8)	S(4) = S(3) = S(1'')	57.7 (2)
S(3) = S(4)	3.948(9)	S(1'') - S(4) - S(2)	62.2 (2)
5(5) 5(1)	5 7 10 (7)	S(2) - S(1'') - S(4)	65.7(2)
		S(4) - S(2) - S(1'')	56.1 (2)
		S(1'')-S(2)-S(3)	60.1 (2)
		S(3) - S(1'') - S(2)	50·5 (2)
		S(2) - S(3) - S(1'')	69.4 (2)
		S(2) - S(4) - S(3)	48.5 (2)
		S(3) - S(2) - S(4)	66.8 (2)
		$\tilde{S}(4) - \tilde{S}(3) - \tilde{S}(2)$	64.7 (2)
P S(1)	2.048(10)	S(1) - P - S(2)	113.4 (4)
$P_{S(2)}$	2.044(10)	S(1) - P - S(3)	112.3 (4)
P - S(3)	2.059 (6)	S(1) - P - S(4')	99.1 (4)
$P_{S(4')}$	2.047(8)	S(2) - P - S(3)	103.4 (3)
S(1) - S(2)	3.420 (9)	S(2) - P - S(4')	114.1 (3)
S(1) - S(3)	3.411 (8)	S(3) - P - S(4')	115.1 (4)
S(1) - S(4')	3.117 (10)	S(1) - S(2) - S(4')	54.1 (2)
S(2) - S(3)	3.218 (8)	S(4') - S(1) - S(2)	63.1 (2)
S(2) - S(4')	3.433 (9)	S(2) - S(4') - S(1)	62.8 (2)
S(3) - S(4')	3.464 (8)	S(1) - S(3) - S(4')	53·9 (2)
		S(4')-S(1)-S(3)	63.9 (2)
		S(3) - S(4') - S(1)	62·2 (2)
		S(1) - S(2) - S(3)	61.7 (2)
		S(3) - S(1) - S(2)	62.1 (2)
		S(2) - S(3) - S(1)	62.1 (2)
		S(2) - S(4') - S(3)	55.6 (2)
		S(3) - S(2) - S(4')	62.7 (2)
GaP	2.892 (7)	S(4') - S(3) - S(2)	61.7 (2)

Assuming pure ionic bonding (with ionic radii for  $S^{2-}$ ,  $Ga^{3+}$ ,  $P^{5+}$ : 1.85, 0.62, 0.35 Å, respectively) the following interatomic distances would result: Ga-S 2.47, P-S 2.20 and S-S 3.70 Å. The observed Ga-S and P-S distances however are both considerably shorter than these (Table 4). The sulphur ions thus seem to be strongly polarized by Ga and especially by P. All P-S distances in the P-tetrahedra are equal within the limits of the standard deviations. In the Ga-tetrahedra differences exist between the Ga-S distances, those pointing to the atoms shared with the P-tetrahedron being somewhat larger than those pointing to the remaining ones. The S-S distances on the other hand are considerably shortened in the P-tetrahedra. In the Ga-tetrahedra two S-S distances are about 3.70 Å (sum of ionic radii), three are remarkably longer, and only the S-S distance corresponding to the common edge with P-tetrahedra is significantly shorter. This distortion of the coordination tetrahedra can also be seen from the bond angles which vary from 48.5 to  $69.4^{\circ}$  for S-S-S angles (ideal  $60^{\circ}$ ) and from 89.3 to 120.5° for S-cation-S angles (ideal 109.5°). Generally, the Ga-tetrahedra are more distorted than the P-tetrahedra.

The most densely occupied directions are [011], [010] and their symmetry equivalents (see Fig. 1). The most densely occupied planes containing these directions are the tetrahedral faces which lie approximately parallel to  $\{100\}$ ,  $\{011\}$ ,  $\{11\overline{1}\}$ , and  $\{10\overline{2}\}$ . This is reflected by the morphology: small polyhedral crystals show the aboved mentioned planes.

#### Comparison with other M<sup>III</sup> thiophosphates

In Table 5 the crystal data of all known  $M^{111}PS_4$  compounds are listed for comparison.

BPS<sub>4</sub>, AlPS<sub>4</sub> and InPS<sub>4</sub> crystallize in acentric space groups with similar cell dimensions whereas GaPS<sub>4</sub> belongs to a centric space group. By transforming the monoclinic GaPS<sub>4</sub> cell by

$$a' = \frac{1}{2}(a+b+c), \quad b' = \frac{1}{2}(a-b-c), \quad c' = a$$

a triclinic cell commensurate with the other thiophosphates results (Table 5). In this setting the sulphur layers in all the compounds are parallel to (112). Their structures differ only in the cation distributions among the tetrahedral interstices between the sulphur layers. In GaPS<sub>4</sub> only alternating interlayers are occupied by

## Table 5. Crystal data of M<sup>III</sup>PS<sub>4</sub> compounds

			GaPS₄ trans-	
Compound BPS <sub>4</sub>	A1PS <sub>4</sub>	GaPS₄	formed	InPS₄
$\begin{cases} a = 5.60 \text{ Å} \\ b = 5.25 \\ c = 9.04 \end{cases}$	a = 5.61  Å b = 5.67 c = 9.05	a = 8.603  Å b = 7.778 c = 11.858	a' = 5.694  Å b' = a' c' = 8.603	a = 5.60  Å c = 9.02
Unit cell		$\beta = 135.46^{\circ}$	$\alpha' = 86.15^{\circ}$ $\beta' = 89.24$ $\gamma' = \beta'$	
Z=2	Z=2	Z=4	Z=2	Z=2
Space group I 222	P 222	$P 2_1/c$		<i>I</i> 4

cations. In the other compounds all the interlayers are occupied by cations but individual differences in their distributions lead to various symmetries.

We are indebted to Professor R. Nitsche for supplying the crystals and for helpful discussions. All numerical calculations were performed on the Univac-1106 of Rechenzentrum der Universität Freiburg.

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# Stereochemistry of an Unusual Trialkyl Sulfonium Ion. The Crystal Structure of 2,3-Dimethyl-2-butenyl-1,1,2-trimethylpropyl-methylsulfonium 2,4,6-Trinitrobenzenesulfonate

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The nature and crystal structure of the product obtained from the reaction of methylsulfonium 2,4,6trinitrobenzenesulfonate with two molecules of 2,3-dimethyl-2-butene has been determined by X-ray methods and the product was found to be the title trialkyl sulfonium salt. The triclinic (PI) lattice parameters are  $a = 13 \cdot 128 \pm 0.009$ ,  $b = 12 \cdot 100 \pm 0.006$ ,  $c = 7.992 \pm 0.003$  Å,  $\alpha = 93.94 \pm 0.04^{\circ}$ ,  $\beta = 102.00 \pm 0.04^{\circ}$ , and  $\gamma = 74.95 \pm 0.04^{\circ}$ . The structure was solved by the heavy-atom technique and refined by full-matrix least-squares calculations to an R value of 0.057 using 3504 observed reflections measured on a diffractometer. Knowledge of the structure of the trialkyl sulfonium ion has led to a suggested mechanism for its formation. The sulfonium ion is pyramidal and it is distorted from 3m symmetry. The C-S<sup>+</sup> bond distances are 1.799, 1.830 and 1.882 Å and increase with increasing bulkiness of the alkyl group on S<sup>+</sup>. The ortho nitro groups of the trinitrobenzenesulfonate anion are twisted by 54 and 61° to the benzene plane while the para nitro group is twisted by only 2.5°. There are a number of intermolecular contacts shorter than 3.4 Å. The shortest contacts are between the trinitrobenzenesulfonate anions.

#### Introduction

In an attempt to synthesize another example of a stable episulfonium salt such as that obtained from the reaction of the olefin cyclooctene with methyl sulfenium trinitrobenzene sulfonate (Pettit & Helmkamp, 1963), the latter was reacted with excess 2,3-dimethyl-2-butene (Carbin & Helmkamp, 1970). Two molecules of 2,3dimethyl-2-butene reacted to give a crystalline product whose structure resisted spectroscopic analysis (Carbin & Helmkamp, 1970). The present X-ray study has unequivocally established the structure of the product (1) and has also led to the proposal of a probable reaction mechanism for its formation. A paper on the details of the chemical work and a brief report on the X-ray structure have been published elsewhere (Carbin, Helmkamp, Barnes & Sundaralingam, 1972).

#### Experimental

Colorless crystals of (I) grown in nitromethane and ether were supplied by Carbin & Helmkamp. The crystal data are as follows:  $C_{19}H_{29}N_3O_9S_2$ , triclinic,  $P\overline{1}$ , Z=2;  $a=13\cdot128\pm0\cdot009$ ,  $b=12\cdot100\pm0\cdot006$ ,  $c=7\cdot992$  $\pm0\cdot003$  Å,  $\alpha=93\cdot94\pm0\cdot04^\circ$ ,  $\beta=102\cdot00\pm0\cdot04^\circ$ ,  $\gamma=$  $74\cdot95\pm0\cdot04^\circ$  (obtained by least-squares fit of eleven reflections measured on a diffractometer),  $D_{obs}=1\cdot412$ g cm<sup>-3</sup> (by flotation in CCl<sub>4</sub> and C<sub>2</sub>H<sub>5</sub>P<sub>2</sub>O),  $D_{calc}=$  $1\cdot406$  g cm<sup>-3</sup>. The crystal data are consistent with the presence of one formula unit of the complex,  $(C_{13}H_{27}S)^+(C_6H_2N_3O_9S)^-$  in the asymmetric unit of the structure. The calculated linear absorption coefficient for Cu K $\alpha$  radiation is 24·2 cm<sup>-1</sup>.

The crystal used for the data collection was a plate of approximate dimensions  $0.25 \times 0.05 \times 0.7$  mm. The intensities of 3990 independent reflections were measured on a Picker four-circle diffractometer using nickel-

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